The Preparation of ${\rm V_2O_5}$ by Ozone Oxidation of ${\rm VOSO_4}$ Solution for Cathode Active Material of Lithium Secondary Battery

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By oxidizing an aqueous solution of $VOSO_4$ by using ozone, yellowish black, powder-like deposit was obtained, the composition of which coincided with V_2O_5 . This deposit had the intermediate structure between crystalline and amorphous V_2O_5 and gave a high capacity of 175 Ah/kg as a cathode active material in a Li secondary battery.

High energy density cathode active materials for rechargeable lithium batteries are earnestly being developed. Vanadium pentoxide(V_2O_5) seems to be one of the most promising materials among the candidates. By using crystalline $V_2O_5(c-V_2O_5)^{1}$ and amorphous $V_2O_5-P_2O_5(a-V_2O_5-P_2O_5)$, 2 a coin type lithium secondary battery was developed and commercialized. In this paper, we present a new preparation method of V_2O_5 , i.e., ozone oxidation method. The obtained material $(0_3-V_2O_5)$ showed a more revised discharge profile than c-V₂O₅ and gave a high capacity of 175 Ah/kg, whose characteristics were almost the same as those of electrolytically prepared V_2O_5 (e- ${\rm V_2O_5})$ from ${\rm VOSO_4}$ solution. $^{3)}$ The ${\rm O_3-}$ and ${\rm e-V_2O_5}$ have the intermediate structure between $c-V_2O_5$ and $a-V_2O_5-P_2O_5$. However, the ozone oxidation method is more efficient than the electrochemical method and is thought to be more useful. To compare the characteristics of $0_3-V_20_5$ with other types of V_2O_5 , $e^{-V_2O_5^{4}}$ and $a^{-V_2O_5-P_2O_5}$, which has a nominal composition of 95 mol% V_2O_5 , are prepared according to a previous paper. The $c-V_2O_5$ is a reagent grade commercialized product(Wakojunyaku Co.). The preparation of $\mathrm{V}_2\mathrm{O}_5$ by the ozone oxidation method is as follows. By bubbling ozone gas prepared using ozonizer(Nihon Ozone Co. 0-1-2) at the rate of 1.5 g/h into acidic 0.2 $\mathrm{mol/dm}^3$ VOSO_4 solution(250 ml) at various pHs and temperatures, a yellowish-black precipitate was obtained. The vanadium content in the yellowish-black precipitate obtained at pH 1.9 was analyzed using an atomic absorption method and was found to coincide with the theoretical value for V_2O_5 . By increasing the pH value, the production rate of the precipitate increased(Fig. 1). If the oxidation reaction proceeds as in the following reaction,

 $2VO^{2+} + O_3 + 2H_2O = V_2O_5 + O_2 + 4H^+$ this tendency can be qualitatively explained. Production rate depends on the reaction temperature(Fig. 2) and suddenly increased at about 40 In order to confirm the oxidation state of precipitate, an XPS analysis was carried out using a Shimadzu ASIX-1000 on the sample obtained at pH 2.5 and 80 OC and on other types of V2O5s(Table 1). All binding energy calibrations were made against Ag $3d_{5/2}$ at 368.0 eV. The surface charging was corrected by measuring the shift of the adventitious carbon(1s) line at 285.0 eV. Argon ions of 2 kV energy were used for ion bombardment of sample surfaces. These binding energies are nearly equal to each other except for $a-V_2O_5-P_2O_5$.

Figure 3 shows the x-ray diffraction patterns of various types of $\rm V_2O_5$, where A is a diffractogram of a-V₂O₅-P₂O₅ showing only a diffused, broad halo characteristic of an amorphous state. Diffractogram C is that of the ozone oxidation product and is almost the same as that of e-V₂O₅(B). Both diffracto-

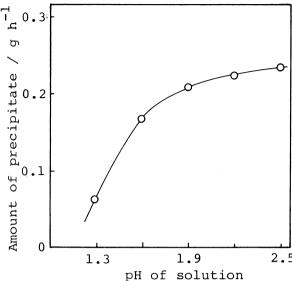


Fig. 1. Relationship between production rate and pH of the solution.

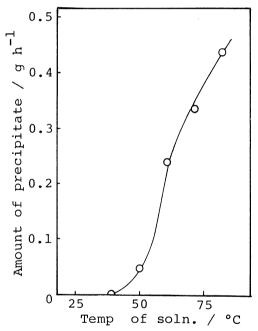


Fig. 2. Relationship between production rate by ozone oxidation and temperature of the solution.

grams show small diffraction peaks at 25.7° which seems to correspond to the 110 diffraction peak and halos. By heating the ozone oxidation product at 400 $^{\rm O}$ C, x-ray diffractogram D is obtained, which is similar to that of c-V₂O₅(E). Thus, the ozone oxidation product(C) is considered to be V₂O₅ (O₃-V₂O₅). The structure of O₃- and e-V₂O₅ is intermediate between a- and c-V₂O₅ and seems to be more like an amorphous material rather than a crys-

talline one. The IR spectra of the five different V_2O_5 s are shown in Fig. 4. The spectrum of $c-V_2O_5$ coincides with that re-

Table 1. Binding Energy of Various Types of V_2O_5

	Binding Energy / eV			
	0 1s	V 3p	V 2p _{3/2}	V 2p _{1/2}
$o_3 - v_2 o_5^{a}$	531.8	41.2	517.0	523.8
$e^{-V}2^{O}5$	531.4	41.4	517.0	523.4
$a-V_2O_5-P_2O_5$	528.4	39.0	514.6	521.4
c-V ₂ O ₅	532.0	42.8	518.0	524.8

a) Obtained in pH 2.5 solution at 80 $^{\rm O}{\rm C}$.

ported by Sanchez and Livage.⁵⁾ The spectrum of 0_3 - V_20_5 heated at 400 O C (D) shows a similar spectrum to $c-V_20_5(E)$. The 0_3 -(C) and $e-V_20_5(E)$ had rather broadened absorption bands compared to those of $c-V_20_5(E)$ showing that the system changed to a random structure with a wide distribution of the V-O bond length and angles like $a-V_20_5-P_20_5(A)$.⁶⁾

Figure 5 shows the charge-discharge curves of the various types of $\rm V_2O_5$ for the second cycle obtained at a constant current of 0.5 mA/cm² from 3.5 V to 2.0 V in a 1M LiClO $_4$ propylene carbonate(PC) solution. Cathode mixtures were fabricated by mixing $\rm V_2O_5$ powder, graphite powder(Wakojunyaku

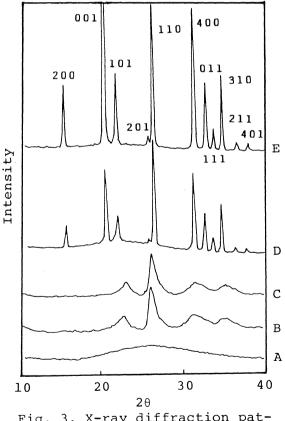


Fig. 3. X-ray diffraction patterns of (A)a- V_2O_5 - P_2O_5 , (B)e- V_2O_5 , (C) O_3 - V_2O_5 , (D) O_3 - V_2O_5 heated at 400°C, and (E)c- V_2O_5 .

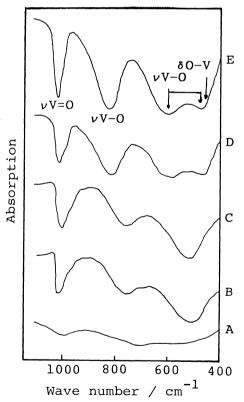
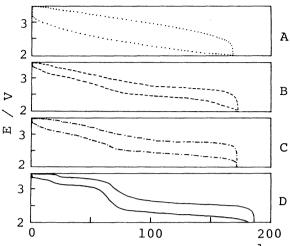


Fig. 4. IR-spectra of (A)a- $V_2O_5-P_2O_5$, (B)e- V_2O_5 , (C)O₃- V_2O_5 , (D)O₃- V_2O_5 heated at 400°C, and (E)c- V_2O_5 .

Co.) and teflonized acetylene black powder(TAB-2, IBA Sample No.2) and pressing the mixture onto a nickel mesh collector at a pressure of 3 tons/cm² to make a 3 cm² tablet. C-V₂O₅(D) showed two small steps at about 3.3 V and 2.3 V and one definite step at about 3 V, whereas $a-V_2O_5-P_2O_5(A)$ showed a smooth discharge curve. On the other hand, $O_3-(C)$ and $e-V_2O_5(B)$ showed almost the same shape and intermediate behavior between a-V2O5-P2O5(A) and $c-V_2O_5(D)$, i.e., two loose steps appeared at about 2.8 V and 2.3 V. The shape of the discharge profile is closely related with the struc-



Specific capacity / Ah kg⁻¹ Fig. 5. Charge-discharge profiles of $(A)a-V_2O_5-P_2O_5$, $(B)e-V_2O_5$, $(C)O_3-V_2O_5$, and $(D)c-V_2O_5$ obtained at 0.5 mA / cm².

ture of $\rm V_2O_5$, which provides a site for lithium intercalation. The x-ray and IR data suggested that the crystal structure of $\rm O_3$ - and $\rm e^{-V_2O_5}$ is not so clear when compared to that of $\rm c^{-V_2O_5}$. The site of lithium intercalation is estimated to be not fixed so rigidly compared to $\rm c^{-V_2O_5}$, which reflects the profile showing loose steps. The cyclability of $\rm O_3$ - and $\rm e^{-V_2O_5}$ was very good. The discharge capacity from the second cycle was about 175 Ah/kg, a little larger than a $\rm V_2O_5$ -P $_2O_5$ and was almost constant, whereas the capacity of $\rm c^{-V_2O_5}$ decreased with cycling.

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